The Crystal Structure of CuVO₃(II), a High-Pressure Ilmenite Phase

J. R. REA* AND E. KOSTINER[†]

Baker Laboratory of Chemistry, Cornell University, Ithaca, New York 14850

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The crystal structure of a second high-pressure copper vanadate phase, CuVO₃(II), has been determined and refined by full-matrix least-squares procedures using automatic diffractometer data to a residual R = 0.042 ($R_w = 0.051$). The space group is rhombohedral, $R\overline{3}$, with hexagonal unit cell a = 4.966(2)and c = 14.084(5) Å $[a_R = 5.501(2)$ Å and $\alpha = 53.66(3)^\circ]$. The structure is the fully ordered ilmenite-type and, on the basis of published magnetic data and the interatomic distances, the valence distribution $Cu^+V^{5+}O_3$ is proposed. This represents a unique example of Cu^+ in an octahedral environment.

Introduction

Chamberland (1) suggested that the two highpressure phases of copper vanadate, $CuVO_3(I)$ and CuVO₃(II), prepared by him were related to the rhombohedral ilmenite structure-type because of the similarities in the powder X-ray diffraction patterns among the three compounds. We have recently confirmed by a single-crystal X-ray structural analysis (2) that $CuVO_3(I)$ is a triclinic distortion of the ilmenite structuretype with ionic distribution $Cu^{2+}V^{4+}O_3$. In this paper we report the results of a similar investigation of $CuVO_3(II)$, which is found to have the undistorted ordered ilmenite structure with Cu⁺ in an octahedral environment.

Experimental Methods

Small single crystals of $CuVO_3(II)$ were generously supplied by Professor B. L. Chamberland. Their preparation at 65 kbar and 900- 1000° C has been described (1). An irregularly shaped crystal was selected; precession photographs confirmed the symmetry to be rhombohedral.

The lattice parameters were determined in a PICK-II least-squares refinement program,

* Present address: Mary Baldwin College, Staunton, VA.

using 34 reflections within the angular range $11^{\circ} < |2\theta| < 29^{\circ}$; the reflections were manually centered on a Picker FACS-I four-circle diffractometer using MoK α radiation. At 22°C the lattice parameters for the triply primitive hexagonal unit cell are a = 4.966(2) and c = 14.084(5) Å; the parameters for the rhombohedral unit cell are a = 5.501(2) Å and $\alpha = 53.66(3)^{\circ}$. The numbers in parentheses represent the standard deviations in the last reported figure. These parameters are in substantial agreement with those derived from the refined powder diffraction data (1). The calculated density is 5.381 g/cm³ ($V_R = 100.3 \text{ Å}^3$).

Diffraction intensities were measured using Zr-filtered MoK α radiation at a takeoff angle of 2.5° with the diffractometer operating in the θ -2 θ scan mode. Scans were made at 1° per min over 1.5° with allowance for dispersion, and with 40-sec background counts taken at both ends of the scan. Of the 309 independent data investigated in the angular range $2\theta < 71^\circ$, a total of 298 were considered observable according to the criterion $|F_o| > 0.675\sigma_F$, where σ_F is defined as $0.02|F_o| + [C + k^2 B]^{1/2}/2|F_o|Lp$; the total scan count is C, k is the ratio of scanning time to the total background count time, and B is the total background count. Three reflections were systematically monitored, and no random variations in intensity greater than 3.5 % were observed over the entire data collection period; the mean variation was very much smaller. However,

[†] Author to whom correspondence should be addressed at Department of Chemistry and Institute of Materials Science, University of Connecticut, Storrs, CT 06268. Copyright © 1973 by Academic Press, Inc. All rights of reproduction in any form reserved.

TABLE I

OBSERVED AND CALCULATED STRUCTURE FACTORS

FCAL	13	8.15	314	983	82	411	32	1358	323	899	116	281	85	1036	337	746	249	157	41	19	131	432	169	637	111	66	137	144	354	294	882	190	1107	383	336	1494	182	1263	Ø	167	142	1378
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two scale factors were used in the refinements, since portions of the data were taken at two different times.

The intensity data were corrected for Lorentz and polarization effects, and absorption corrections were made. Since the crystal was irregularly shaped, the absorption corrections were made using a computer program written by N. W. Alcock and modified by B. Lee for a crystal of general shape. Input for the program included the indices of the plane faces of the crystal, which were closely approximated with the aid of precession photographs, and the orientation angles of the crystal coordinate system with respect to the diffractometer coordinate system, which were obtained from the program PICK-II. The maximum relative absorption correction applied was 13% of $|F_a|$.

Determination and Refinement of the Structure

The space group of CuVO₃(II) was reported to be $R\bar{3}m(1)$. However, precession photographs were carefully examined, and the true Laue symmetry was found to be $\bar{3}$, allowing the possible space groups R3 or $R\bar{3}$. Analysis of the Patterson map revealed more than one possible set of positions for the copper and vanadium atoms on the hexagonal *c*-axis. The best set was determined by a series of least-squares refinements, and a difference Fourier map clearly revealed the oxygen sites.

Three cycles of least-squares refinement (4) in the space group $R\overline{3}$ (the space group of the ilmenite structure-type), using a $1/\sigma^2$ weighting scheme, zerovalent scattering factors for Cu, V, and O (5), isotropic temperature factors, and corrections for secondary extinction and anomalous dispersion yielded a residual R = 0.054and a weighted residual $R_w = 0.065$. Since CuVO₃(II) was reported to be nonstoichiometric (1), an anisotropic refinement was run in which the occupancies of the vanadium and oxygen sites were allowed to vary. The occupancies did not deviate significantly from 100%, and R was not lower than R for fixed occupancies. Furthermore, the final difference Fourier map was flat, except for minor background. This indicates a fully ordered ilmenite structure. The final anisotropic refinement (occupancies fixed at 100%), based on a data:parameter ratio of 16.5 with 18 independently varied parameters, yielded a final R = 0.042 and $R_w = 0.051$ for the observed data. Calculated and observed structure factors are listed in Table I. In the final refinement the maximum extinction correction (6) was 7% of $|F_c|$ for the 104 reflection.

Table II presents the final atomic coordinates and anisotropic thermal parameters.

Discussion

 $CuVO_3(II)$ has the ordered ilmenite structure. The ilmenite structure can be described (7) as a hexagonal close-packed array of oxygen ions with metal ions occupying two-thirds of the octahedral holes. The different metal ions are in alternating layers such that each octahedron containing a copper ion shares a face with a vanadium octahedron above or below it (but not both) and also share edges with three other copper octahedra in the same layer. Vanadium octahedra have similar linkages. This means that the octahedral holes of the oxygen array, which are linked by face-sharing into chains parallel to the hexagonal c-axis, are filled in the sequence vacancy-Cu-V-vacancy-V-Cu-vacancy, etc. As in CuVO₃(I), the unlike metal ions in the face-sharing octahedra repel each other so that they are further apart (2.998(2))Å) than the like metals of edge-sharing octahedra within a layer (2.951(1) Å for Cu; 2.930(1))Å for V).

TABLE II

FRACTIONAL ATOMIC COORDINATES AND ANISOTROPIC THERMAL PARAMETERS⁴

Atom	10 ⁴ x	10⁴ <i>y</i>	10 ⁴ z	B ₁₁	B ₂₂	B ₃₃	B ₁₂	B ₁₃	B ₂₃
Cu	0	0	1418.3(5)	0.59(2)	B ₁₁	1.02(3)	$\frac{1}{2}B_{11}$	0	0
v	0	0	3547.1(6)	0.44(2)	B_{11}	0.65(3)	$\frac{1}{2}B_{11}$	0	0
0	3704(5)	80(5)	765(1)	0.65(7)	0.43(7)	0.74(7)	0.27(6)	0.12(5)	0.04(5)

" Numbers in parentheses are estimated standard deviations in the last significant figure.

TABLE	111
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	Distance (Å)	Angle (°)	Edge length (Å)
i. Copper polyhedron ^b			
3x Cu–O	2.039(2)		
3x Cu–O'	2.241(2)		
3x O-Cu-O		101.2(1)	3.152(4)
3x O-Cu-O'		93.0(1)	3.107(4)
3x O-Cu-O'		89.2(1)	3.008(3)
3x O'-Cu-O'		73.5(1)	2.682(4)
3x O-Cu-O'		160.3(1)	4.216(2)
ii. Vanadium polyhedron ^b			
3x V-O	1,788(2)		
3x V-O'	2.073(2)		
3x O-V-O		102.6(1)	2.790(4)
3x O-V-O'		81.6(1)	2.531(4)
3x O-V-O'		93.6(1)	2.820(4)
3x O'-V-O'		80.6(1)	2.682(4)
3x O-V-O'		161.9(1)	3.814(3)
iii. Oxygen polyhedron			
V-O-Cu		120.9(1)	3.331(1)
V-O-V'		98.4(1)	2.930(1)
V-O-Cu'		135.8(1)	3.737(1)
Cu–O–V′		126.7(1)	3.675(1)
Cu-O-Cu'		87.0(1)	2.951(1)
V'-O-Cu'		88.0(1)	2.998(2)
iv. Metal-metal distances			
3x CuCu' (in plane)	2.951(1)		
Cu'-V' (face sharing)	2.998(2)		
3x Cu-V (edge sharing)	3.331(1)		
3x V-V' (in plane)	2.930(1)		

BOND DISTANCES	BOND ANGLES	AND POLYHEDRAL	EDGE LENGTHS ⁴
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"Numbers in parentheses are estimated standard deviations in the last figure.

^b The primed and unprimed oxygen atoms belong to different close-packed layers.

The fact that $CuVO_3(II)$ exhibits Van Vleck paramagnetism (1) implies that the molecular species in their ground states have no paramagnetic moment. Therefore, Cu and V in this compound exist in d^{10} and d^0 configurations, respectively; that is, the valence distribution is $Cu^+V^{5+}O_3$. Analysis of bond distances supports this argument to some extent. Table III lists bond distances, angles, and polyhedral edge lengths. The average Cu–O and V–O bond lengths are 2.140 and 1.931 Å, respectively. Using the ionic radius for four-coordinated oxygen (1.38 Å) derived by Shannon and Prewitt (8), the ionic radii for copper and vanadium in CuVO₃(II) are 0.76 and 0.55 Å, respectively. For $Cu^{2+}V^{4+}O_3(I)$ the ionic radii for Cu and V were found to be 0.75 and 0.56, respectively. Although the derived ionic radii in these two compounds are not markedly different, we believe, considering the accuracy of the data, that this slight difference supports the lower valence state for copper (larger ionic size) and higher valence state for vanadium (smaller size) in CuVO₃(II)—that is, the $d^{10}-d^0$ configuration implied by the magnetic data.

The formation of the ilmenite-related vanadate phases is apparently related to the valence state of the vanadium involved. Each of the four triclinic high pressure phases $MVO_3(I)$ (M = Co, Cu, Ni, Mn) contains vanadium as V⁴⁺ (1, 2, 9, 10). The only one of these which forms a second (undistorted) high pressure ilmenite phase is $CuVO_3$, in which the copper atom can exist as a stable univalent cation thereby allowing vanadium to occur as V⁵⁺, and which seems to have the unique feature of Cu⁺ in an octahedral site.

Acknowledgments

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References

I. B. L. CHAMBERLAND, *J. Solid State Chem.* 1, 138 (1970).

- 2. J. R. REA, P. W. BLESS AND E. KOSTINER, J. Solid State Chem. 5, 446 (1972).
- "International Tables for X-Ray Crystallography", Mathematical Tables, Vol. 2, p. 295. Kynoch, Birmingham, England (1968).
- W. R. BUSING, K. O. MARTIN AND H. A. LEVY, Oak Ridge National Laboratory, ORNL-TM-305. Oak Ridge, TN (1962).
- 5. D. T. CROMER AND J. B. MANN, Acta Crystallogr., Sect. A 24, 321 (1968).
- W. H. ZACHARIASEN, Acta Crystallogr. 23, 558 (1967); Sect. A 24, 324 (1968).
- 7. G. SHIRANE, S. J. PICKART, R. NATHANS AND Y. ISHIKAWA, J. Phys. Chem. Solids 10, 35 (1959).
- R. D. SHANNON AND C. T. PREWITT, Acta Crystallogr., Sect. B 25, 925 (1969).
- 9. B. L. CHAMBERLAND, J. Solid State Chem. 2, 521 (1970).
- 10. Y. SYONO, S. AKIMOTO AND Y. ENDOH, J. Phys. Chem. Solids 32, 243 (1971).